Development of Colloidal Semiconductor Nanocrystals: Synthesis, Properties and their Outlook for Light Emitting Diodes (LEDs)

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ABSTRACT: In recent years, the optimum properties of colloidal nanocrystals have widespread its usage in electronics and optoelectronics applications. The advancement in synthesis enables control over the size, shape, crystal structure and composition of nanocrystals. It facilitates control over optical and physical properties of colloidal nanocrystals. They have tremendous potential to be used in different applications. This review focuses on recent progress in the study of colloidal semiconductor nanocrystals. Here, we review the advances in the synthesis techniques, optical properties, surface modification with ligand treatment and doping in colloidal semiconductor nanocrystals. This enables their use in various applications such as light emitting diodes, lasers, solar cells, biomedical labelling and gas sensors. Literature studies reveal that the incorporation of any impurity element or defect into semiconductor affects its optical properties as well other physical properties. Binary semiconductor nanocrystals from different groups have high degree of ionicity making them suitable for various applications. With the advancement in technology, ternary semiconductor nanocrystals have replaced the binary nanocrystals as they possess tunable emission in the visible region of spectrum. In this review, we have summarized different ternary semiconductor nanocrystals which have been applied in light emitting diodes, biosensor, imaging, solar cell and drug delivery.

Keywords: Semiconductor nanocrystals, optical properties, binary nanocrystals, ternary nanocrystals, tunable emission

I. INTRODUCTION

Light emitting diodes (LEDs) have drawn considerable concern in research field as well in industries due to its energy saving potential. Their remarkable characteristics make them efficient for use in artificial lighting and displays. Now a days, white light emitting diodes (WLEDs) have been attained by integrating color converting material with commercial available blue LEDs. Previously, phosphor doped rare earth elements have been used along with blue LEDs to generate WLEDs which were inefficient. Semiconductor nanocrystal quantum dots exhibit the potential to replace the rare earth phosphors as they possess high color rendering index (CRI), high luminous efficacy (LE), solution processibility, tunable color and their raw material is abundant in nature. WLEDs have replaced traditional light emitting systems (incandescent bulbs and CFLs) as they possess long lifetime and great economic significance. In order to achieve the full potential of WLEDs, advanced technologies must be developed to produce a high quality white light that is highly efficient, has a long lifetime, and is cost competitive. So the ideal WLEDs should emit a broad spectrum of white light that is similar to natural sunlight and most appealing to the human eyes. Based on the recent developments in semiconductor nanocrystals-based solid-state lighting (WLEDs), three different approaches are used to generate white light i.e. discrete color mixing, color conversion, and direct white light generation. The WLEDs first generated in 1996 were phosphor based. Semiconductor nanocrystal quantum dots exhibit the potential to replace the rare earth phosphors as they possess high CRI, high LE, solution processibility, tunable color and their raw material is abundant in nature. As ideal white LEDs (WLEDs) should be highly efficient, cost competitive, having a long lifetime, hence high-efficiency light conversion materials are intensively required. The quality of white light generated from WLEDs can be evaluated by using several parameters, such as Commission Internationale de l’Eclairage (CIE) chromaticity coordinates, color rendering index (CRI), correlated color temperature (CCT), and luminous efficacy (LE). The color rendering index mainly tests the color rendering ability of any test light source with respect to a reference light source. The reference source in such cases is assumed to possess perfect color rendition. The best CRI is considered as 100 whereas -100 is the representation of the worst color rendition. The CQS is the improved parameter as compare to CRI which is used to calculate the performance of the final color rendering of any light source with the scale lies between 0 to 100. Another parameter is correlated color temperature which defines the temperature of light source being tested. The warmer white light possess CCT value between 3000 and 4500 K is mostly desirable for use indoor lighting applications. This review will focus on the development
of the semiconductor nanomaterials and their use in Semiconductor nanocrystals (NCs) exhibiting size dependent optical and electronic properties have been intensely investigated and developed for use in diverse applications i.e. optoelectronic devices, solar energy conversion, bioimaging and targeted drug delivery [1,2]. Quantum dots (QDs) are those three-dimensionally confined nano-crystalline semiconductor materials whose radius is equal to or less than the exciton bohr radius. Thus, quantum dots exhibit distinctive properties as a function of its size. The confined exciton in semiconductor precisely tune the band gap energy which depends upon the dimensionality and degree of confinement [3]. As emicoductor QD comprises atoms of group II–VI elements (e.g., CdTe, CdSe, CdS, ZnS, ZnSe, or ZnTe), group III–V elements (e.g., InP or InAs), group IV–VI elements (e.g., PbSe, PbS, or PbTe), or group IV elements (e.g., Si, C, or Ge). Doping of transition elements is an important aspect to tune the properties of colloidal nanocrystals. The major problem in undoped semiconductor nanocrystals is the self-absorption. Therefore, transition metal ions-doped semiconductor NCs are increasingly being investigated. Doping of an impurity atom to the host matrix perturb the crystal structure by creating local quantum states within the band gap of the material. Various transition atoms have been doped into the host matrix including Mn, Cu, Ag, Fe, Co etc. It is the amount of impurity element and its position in the structure that effects the properties of the nanocrystals. Along with maintaining the intrinsic advantages of QDs, the doped nanocrystals exhibit additional merits such as, large stokes shift, enhanced chemical and thermal stability, minimizes the issue of self-absorbance and longer state lifetime [4-5]. The main feature of dopant emission is that the energy is red shifted with respect to band gap energy of the host material which leads to overcome the problem of self-absorbance of dopant emission by host. The dopant or transition metal ions introduce discrete energy states or deep trap levels within the band gap of host semiconductor, which act as luminescent centers. As a result, they show different emission mechanisms as depicted in Fig. 1. For example, Mn$^{2+}$ doped nanocrystals show emission in yellow-orange spectral window (580-600 nm) due to $^4T_{1g}^{2-}A_{1g}$ transition [6-7] and Cu doped nanocrystals shows tunable emission over the visible region that depends on the nature, composition and size of host nanocrystals [8].

![Conduction Band (CB)](image)

Fig. 1. Mechanism of emission in Cu- and Mn-doped colloidal nanocrystals.

In comparison to the binary nanocrystals, the ternary nanocrystals including I–III–VI (I = Cu, Ag; III = In, Sn, Ga, Al; VI = S, Se, Te etc.), or II–III–VI (II = Zn; III = Al, Ga or In; VI = S, Se or Te) chalcopryite compounds have been considered as promising candidates to develop tunable emission by varying their composition. As the energy band gap of the alloyed nanocrystals can be engineered by varying the composition of the material [9], they show mixed or intermediate optoelectronic properties. For example, the band gap energy of ZnCdS semiconductor nanocrystals lies between the band gap of ZnS and CdS nanocrystals [10]. Generally, there are some unsatisfied or dangling bonds present at the surface which may be due to non-stoichiometry or voids present in the structure that leads to the trap or surface state defects. These trap states quenches the radiative recombination and decreases the quantum yield (QY). Therefore, it is required to passivate or capped these trap states for the stability of nanocrystals. In order to passivate the surface, inorganic [11] as well as organic [12], capping agents have been used. In case of inorganic agents, a material having higher energy band gap is chosen as a shell to passivate the core material.

The tunable emission covering the whole visible region (UV-vis-NIR), direct band gap, large stokes shift and longer fluorescence decay lifetime are the main features that enables the use of colloidal nanocrystals in energy efficient applications.

It is also notable that various methods had been reported for synthesizing the monodispersed and size controlled nanocrystals with the variation in reaction parameters. The temperature, reaction time and precursor ratio are the main parameters, which have been considered during synthesis of nanocrystals. The main approaches used to synthesize nanocrystals include hot injection method, solvo/hydro thermal method, non-injection method, refluxing, wet chemical synthesis, sol-gel process, micro-emulsion, molecular beam epitaxy (MBE) and physical vapor deposition (PVD) etc. Fig. 2 represents the methods used to synthesize colloidal nanomaterials of different structures.

In hot injection method (Fig. 3), a cool solvent is added into the hot solution of cationic precursors at appropriate high temperature, which is then followed by fast nucleation [13]. This is the most suitable method that allows the synthesis of controlled size nanocrystals. The injection temperature of precursor is the most important parameter as the size and morphology of the resultant nanocrystals depends on it [14]. Another synthesis method is wet chemical technique or aqueous synthesis approach that mainly follows the conventional precipitation process. This is the simplest method used to synthesize nanocrystals [15].

In sol-gel method, a sol of metal precursor is prepared followed by the polymerization to form gel. This method is cost effective, simple and can be used for large scale synthesis [16]. However, the rate of defects present in it is more. Many researchers have reported microemulsion method (reverse micelle) to synthesize nanocrystals [17]. Microreactor synthesis have also been proposed in which a metal precursor is passed to micromixer through microchannels that further passed to number of zones having specific temperature gradient. This method require small amount of precursor and short reaction time [18].

**Fig. 2.** Generic synthesis methods used to synthesize colloidal nanocrystals (CNCs).

**Fig. 3.** Hot-injection method for the synthesis of monodisperse nanoparticles. (Adapted from ref 13. Copyright2007 Wiley-VCH Verlag GmbH).
The non-injection method also called heating up method includes the addition of precursors in one pot followed by rapid heating to specific temperature at which nucleation occurs (Fig. 4) [16]. This is also a best method for large scale synthesis [17]. Solvothermal method is used to synthesize nanocrystals by using autoclave. The anionic and cationic precursors are added at low temperature and the sealed autoclave is heated to desired temperature [18]. Thermolysis is a fast synthetic process where metal precursors are heated at high temperature. During the synthesis of nanocrystals, the reaction parameters play a significant role for getting nanocrystals of desired shape and size. The change in precursor ratio has a significant effect on the optical properties. The reaction time and temperature are crucial parameters as the prolonged reaction time leads to aggregation and effect the stability of nanocrystals. The high temperature and more reaction time results precipitation of the nanocrystals. The capping ligand used and the pH of the reaction also affects the optical properties. Afterwards, these high quality nanocrystals can be used in various applications. Sometimes, their hydrophobic nature makes them unsuitable for use in some applications. So, surface modification is required to change their medium from oil based to water soluble. Ligand exchange and encapsulation of the nanocrystals are the main approaches used by researchers for the surface modification of nanocrystals. They improve their water solubility and make them efficient for use in biological applications. Unique properties of nanocrystals makes them effective for use in various applications like optoelectronics, photonics and biomedicine. Some of them are represented in Fig. 5.

![Image](image-url)

**Fig. 4.** Noninjection heat-up method. All precursors are present from the beginning of the reaction, and are steadily heated to initiate nucleation and growth stages. (Adapted from ref 16. Copyright 2007 American Chemical Society).

![Image](image-url)

**Fig. 5.** Illustrates various applications of colloidal nanocrystals.
II. DEVELOPMENT OF SEMICONDUCTOR NANOCRYSTALS (NCs)

A. Binary semiconductor nanocrystals

The unique optical properties of the colloidal nanocrystals make them proficient as they exhibit bandgap tunability, wide absorption range, high luminescence, spectral purity and photo-/chemical stability [19]. Their novel applications in various fields such as light emitting diodes (LEDs), laser diodes, integrated circuits, photodetectors have generated their technological interest. The most common binary semiconductor compounds consisting of II-VI elements, include cadmium sulfide (CdS), cadmium selenide (CdSe), cadmium telluride (CdTe), zinc selenide (ZnSe) and mercury sulfide (HgS), etc. These materials mostly reveal wurzite and zinc blende tetrahedral crystal structures. II-VI semiconductors nanocrystals show emission in blue-green spectral region and have been used as blue lasing material in optical wave guide [20].

The oxides from II-VI group like ZnO has been used in biological [21]. There are some QDs comprising III-V elements, i.e., indium phosphide (InP), gallium nitride (GaN), indium arsenide (InAs), etc. These materials have direct band gap, high refractive index and possess tetrahedral crystal structure. Therefore, they have been considered as proficient materials for optoelectronics and photovoltaic applications. These materials have been used in fabrication of high efficiency solar cells [22]. Various nitrides from III-V group such as GaN, InN and AlN possess high thermal conductivities and electron transport properties, which make them efficient for use in optoelectronic properties.

(a) Binary Undoped semiconductor nanocrystals

Among II–VI semiconductors, ZnS, CdS, ZnO, CdTe, etc. are the main candidates as they exhibit favorable electronic and optical properties for optoelectronic applications. Among them, ZnS is a commercially important semiconductor having a wide optical band gap, which makes it very attractive material for optical application especially in nanocrystalline form. The luminescent material ZnS was developed by Theodor Sidot in 1866 [23]. Murray, Norris and Bawendi [24] were the first who reported high-temperature colloidal synthesis of monodispersed CdSe, CdS and CdTe colloidal NCs of uniform size via the hot-injection technique in 1993. Cho et al. [25] synthesized PbSe nanowires and nanorings by adjusting the reaction conditions (i.e. temperatures of 190-250°C and mixture of surfactants). They have successfully synthesized straight, zigzag, helical, branched, and tapered nanowires as well as single-crystal nanorings by using one-pot reactions in a controlled way by careful adjusting the reaction conditions (Fig. 6).

The II–VI binary systems including CdSe, CdS, and PbS magic-sized or regular sized quantum dots (MSQDs) have gained more interest in recent years. The solvothermal synthesis of CdS, CdSe and CdSe/CdS core/shell, PbSe and InP quantum dots in organic solvents had been reported [15, 26-31].

![Fig. 6. (a) Octahedral shape PbSe nanocrystals, (b) Star-shape PbSe nanocrystals and (c) Helical nanowires formed upon annealing of straight PbSe nanowires. Reprinted with permission from ref 25. Copyright 2005 American Chemical Society.](image-url)
It required adequate temperature and pressure which provide the suitable condition for the reaction to takes place for the growth of nanocrystals. The nanocrystals of different sizes have been synthesized by changing the concentration of ligands in noncoordinating solvents. Another approach used for the synthesis of quantum dots is aqueous route that is simple and cost effective. The quantum dots synthesized by this method are attractive for biological applications due to their water dispersibility property. Various materials including CdSe, CdTe, CdTe/ZnTe, CdTe/CdS, CdTe/CdSe, ZnS and ZnSe had been synthesized by this method [32]. Moreover, the most popular stabilizers used for aqueous synthesis of quantum dots are thiols and thioalkyl acids which includes thioglycolic acid (TGA) and 3-mercaptopropionic acid (MPA). It has been found that the synthesized QDs are homogeneous, spherical, and small but their quantum yield is low.

The high-quality quantum dots had been synthesized by hydrothermal method including CdTe, CdTe/CdS core/shell, CdTe/CdSe core/shell, and CdTeS alloyed [33-36]. In this, the precursors containing anions and cations react at high temperature followed the nucleation and growth stages. An another approach that had also been used to produce high-quality colloidal semiconductor nanocrystals both in organic as well in aqueous media is microwave-assisted method. The materials synthesized by this method include CdTe, CdTe/CdS/ZnS, CdSe, ZnSe(S) alloyed, and CdSe(S) alloyed [37-41].

Peng et al. used Cd-phosphonate or Cdcarboxylate complexes in place of dimethyl cadmium which were previously used by Murray et al. to obtain tunable emission ranging between blue to the near UV region [42]. The same group then replaced the coordinating solvent TOPO by the non-coordinating 1-octadecene (ODE) and obtain monodispersed CdS NCs [28]. Henglein et al. coated CdS particles with Cd(OH)2 and increased the quantum yield to 50% by eliminating the trap state emission which was due to high surface to volume ratio [43]. Hines and Guyot-Sionnest [44], synthesized CdSe/ZnS nanoparticles with quantum yield of 50%. ZnS shell growth had been achieved by the injection of a mixture of the organometallic precursors diethylzinc and hexamethyl disilathianeat high temperature (300°C). After that Bawendi et al. synthesized size series of CdSe/ZnS NCs by growing shell at different temperatures of 140–220°C (depending on the core size), and reported their in-depth characterization [45]. A similar approach had been applied for depositing ZnS shell on Cd Se nano rods with lengths up to 30 nm [46]. The addition of hexadecylamine (HDA) to the traditionally used solvent system i.e., trietylphosphine oxide /trietylphosphine(TOPO/TOP) led to abetter control of the growth kinetics during CdSe core and ZnS shell synthesis, results a lower size distribution and increased quantum yields (QYS) of theorder of 60% [21]. An extremely small CdSe/ZnS core shell (CS) NCs had been synthesized by Kudera et al., making blue spectral region accessible with this system [47]. The synthetic approach was based on the sequential growth of Cd Semagic-sized clusters in a mixture of trietylphosphine, dodecylamine, and nonanoic acid ata temperatures of 80°C and their subsequent coating with ZnS. Jun and Jang suggested an another method to access this spectral region [48]. The precursors such as zinc acetate and octanethiol having relatively low reactivity were used to grow ZnS shell at 300°C. The shell material was diffused into the core, resulting in a significant hypsochromic shift of the emission wavelength. The obtained NC gives emission at 470nm with a QY of 60%. Generally, shell growth on wurtzite CdSe core NCs obtained with hot-injection method, is kinetically driven along the c-axis. Therefore, to enhance the growth and uniformity of shell, Lim et al. had used zinc blende CdSe NCs for overgrowth with ZnS and achieved 50% PL efficiency in water [49].

With the advancement in synthesis techniques of colloidal nanostructures, the self assembly of colloidal components had also been extended from micrometer size to nanometer with distinctive intrinsic properties. Chen et al. [50] obtained a superlattice by self-assembly of CdTe and CdSe and estimated the pair interaction energies due to core-core and ligand-ligand vander Waals (V DW) forces. Bang et al. [51] synthesized ZnTe/ZnSe (core/shell) quantum dots (QDs) with PL emission extended from blue to amber (Fig. 7). They have passivated QDs by ZnS layers that maintained their optical and chemical stabilities and demonstrated their potentials of heavy-metal leakage free QDs toward nontion and medical applications. They have also explored their use in photovoltaic applications. Talapin et al. [52] synthesized CdSe NCs with low size distribution by using Hexadecylamine (HDA) in the TOPO/trietylphosphine(TOP) mixed solvent. Ding et al. synthesized water soluble and highly luminescent ZnSe/ZnS core/shell nanocrystals using aqueous method. The tunable emission had been achieved with quantum yield in range of 23% to 62.8% [53]. Fu et al. synthesized CdSe/ZnS/ZnS core/shell nanocrystals [54]. Hu et al. used an effective approach for better quality and stability by using an outer shell that coated the inner core/shell QDs. They had utilized a double outer shell for inner CdSe/CdS core/shell QDs and with an optimal outer shell thickness, the QY was increased to 72% and the QDs show stability even after purification and exposure to UV [55].

Different materials had been used for double outer shells which improved the luminescence efficiency [56]. The device performance of the alloyed core [57] and alloyed intermediate shell [58] had been significantly improved by coating the inner core/shell QD with single outer shell. With optimally graded alloyed intermediate shell, anexternal quantum efficiency (EQE) exceeding 20% and a current efficiency higher than 80 cdA−1 had been achieved [59]. Boonsin et al. investigated the optical properties of gradient alloyed (CdSe)x(ZnS)1−x and (CuInSe2)1−y(ZnS)y QDs. The optical properties of composite films based on these QDs were also studied. The combination of both QDs revealed that it is possible to produce a warm white light by combining with UV commercial LED [60]. Wang et al. synthesized highly luminescent silica coated CdS/CdSe/CdS nanoparticles having quantum yield of 72%. The synthesized nanoparticles having spherical quantum well type structure show excellent stability of silica suggesting their use in real-life applications such as bio-imaging and light emitting devices [61].

(b) Binary Doped semiconductor nanocrystals
The metal ions such as Mn, Cu, Co, Ni and Fe doped nanocrystals had gained significant interest and used for many applications including light emitting diodes and solar cells [62-63].
Fig. 7. Absorption and (b) emission spectra of ZnTe/ZnSe (core/shell) QDs. The inset in panel (b) shows various ZnTe/ZnSe (core/shell) QD samples under UV illumination. (c) Photoluminescence (PL) quantum efficiencies (closed circle, □) and full widths at half-maximum (open squares, □) for ZnTe/ZnSe (core/shell) QDs with 2.2-nm-radius cores and different thicknesses of ZnSe shells. (d) Room-temperature normalized photoluminescence decay of ZnTe/ZnSe QD. (Reprinted with permission from ref 51. Copyright 2010 American Chemical Society).

Fig. 8. “Demonstrate the colloidal nanocrystals covering the whole visible spectrum”. They covered the whole visible spectrum as demonstrated in Fig. 8. Other materials like ZnS:Tb (efficient green emission) [64], ZnS:Sm (weak red emission) [65] and ZnS:Ho (white) [66] had received some interest in later years. Mn-doped ZnS was probably the most studied doped sulfide in nano-sized form. The chemical similarity between Zn$^{2+}$ and Mn$^{2+}$ facilitates the incorporation of the dopant ions into the host crystal. Nevertheless, the incorporated concentration was consistently lower than the intended dopant concentration [67] and part of the Mn$^{2+}$ ions reside near the surface of the nanoparticle. The work by Bol et al. showed that the lifetime of Mn$^{2+}$ in ZnS:Mn$^{2+}$ quantum dots is very similar to the bulk lifetime, in spite
of earlier work stating a considerable shortening of the lifetime [68-69]. Also the emission spectrum itself was almost size-independent [67]. Hence, growing an inorganic (ZnS) shell around ZnS:Mn effectively reduces non-radiative decay paths and enhances the luminescence [70], more efficiently than organic passivation of the surface [67]. Synthesis techniques, luminescence properties and possible applications of Mn-doped ZnS nanoparticles had been explained in detail in the reviews by Chen et al. [71, 67]. The luminescence properties of nano-sized ZnS:Eu had been studied and the results show deviation with the presence of both Eu$^{3+}$ and (assumed) Eu$^{2+}$ emission [72]. Bulk ZnS:Eu was not luminescent and it was proposed that increasing the band gap of the host would allow the 5d-4f transition in Eu$^{2+}$. Incorporation of Eu in ZnS appears difficult as the Zn$^{2+}$ and Eu$^{2+}$ ions possess different ionic radius [73]. Xie et al. [74]. In 2005 synthesized high quality Cu:InP and Cu:InP/ZnSe core/shell quantum dots by varying the size of the InP host nanocrystals for biomedical applications that covered the red and near-infrared (NIR) window (i.e. from 630 to 1100 nm) (Fig. 9). The PL QY of the resulting pure dopant emission i.e. Cu:InP/ZnSe core/shell quantum dots reached close to 40% by the complete elimination of InPb and gap emission.

AC electroluminescence (EL) was reported by using ZnSe/ZnS:Mn/ZnS core/shell nanoparticles having high photoluminescence quantum efficiency of 65% [75]. The semiconductor part of the ACT-FEL devices consisted of a multi-layer of spin-casted nanoparticles layers (thickness of 30 nm) and sputtered (undoped) ZnS layers (12 nm). The typical orange emission (ZnS:Mn) was obtained for these structures having of 2 cd/m$^2$ at 30 kHz. The devices without the sputtered ZnS layers did not show EL emission. Toyama et al. [76] also obtained electroluminescence of ZnS: Mn nanoparticles with a 150 nm thick printed layer and suggested that 5 kHz, and 45 V above threshold were sufficient to obtain a luminance of 1cd/m$^2$ for the smallest particles having size 3 nm. Corrado et al. [77] synthesized Cu-doped ZnS nanocrystals by using mercaptopropionic acid (MPA) as a capping agent. The incorporation of Cu ions into the crystal structure red shifted the emission band in PL spectra. They had suggested that the optical and structural properties make this material as potential candidate for use in solid state lighting, imaging, and other photonic devices. Xin et al. [77] synthesized Cu, Tm-doped ZnS nanocrystals at 200°C by using hydrothermal method. They have found that for long time treatment, it showed low luminous intensity. They have obtained a broad emission band with blue and green emission. Hazarika et al. [79] synthesized Mn-doped ZnS-CdS alloys with narrow energy window ≤150meV in the orange-red emission region. They carried the single crystal PL emission and found that it vary over wide range (~370 meV) which covers the deep green to deep red region of spectrum.

![Fig. 9. PL spectra (a) and PL peak positions (b) of differently sized InP q-dots and Cu: InP d-dots. (Reprinted with permission from ref 74. Copyright 2009 American Chemical Society).](image-url)
Mei et al. [80] synthesized highly luminescent ZnS:Cu, Eu semiconductor nanocrystals by using hydrothermal method and their results demonstrate that this material can be effectively used for optoelectronic applications. Gao et al. [81] fabricated white light emitting diodes by using Mn-doped CdS/ZnS core/shell nanocrystals. They had reported an improvement in the intensity, stability, and achieved color rendering index of 88 and luminous efficiency of 108lm/W. Cooper et al. [82], in 2015 synthesized a series of ZnSe/ZnS QDs by using Cu$^{+}$ as the primary dopant and trivalent cations (Al$^{3+}$, Ga$^{3+}$, and In$^{3+}$) as codopants with tunable PL emission covering 430−570 nm with average full width at half maximum of 80 nm (Fig. 10).

B. Ternary semiconductor nanocrystals

The ternary semiconductor compounds have been widely investigated. The ternary NCs provide tunable emission fromNIR region through visible and covers the ultraviolet (UV) region.

(a) I-III-VI CNCs

Recently, ternary I-III-VI (I=Cu, Ag; III=In, Sn, Ga, Al; VI=S, Se, Te etc.) semiconductor NCs came into view due to their innovative properties. They are less toxic as compared to other Cd based materials. Along with large stokes shift, high photostability, high quantum yield and longer photoluminesence lifetime, they also exhibit wide range of optical and electronic properties which can be achieved by varying their composition and altering structure [83]. As compared to binary CNCs, the ternary I−III−VI compounds have been considered as promising candidates for light emitting and biological applications as they composed of less toxic elements. They have direct band gap in the visible region as CuInSe$_2$ (CISe), CuInS (CIS), AgInS$_2$ (AIS) and AgInSe$_2$ (AIS) have band gap of 1.05 eV, 1.5 eV, 1.87 eV and 1.2 eV, respectively [84]. Various metal salts and sulphur sources had been used to synthesize I-III-VI type CNCs (CIS, AIS, CISe and AISe etc.). Various methods such as electrode position, spray pyrolysis, sputtering, ionized cluster beam technique and chemical vapour deposition had been used to synthesize colloidal CuInS CNCs.

Moreover, composition variation consequences on the defects of I-III-VI QDs can directly tune the optical properties. The presence of different atoms in their composition challenges tuning of the absorption spectra. Several works showed that the non-stoichiometry of ternary QDs could be controlled by the molar ratios of precursors [85]. The tunable emissions are pronounced properties of I-III-VI QDs which attracts various researchers to make their use in light emitting applications. Fabrication of I-III-VI QDs were adapted from the methods used in II−VI binary QDs [13,20]. During synthesis of QDs, the reaction temperature and time [86], injection temperature in hot-injection methods [87], the reactivity and the stoiometric ratios of precursors, the solvent type used, surfactant, and pH [88] etc. are important parameters to adjust the size. A number of metal salts and sulphur precursors were used for the preparation of luminescent I-III-VI QDs by thermolysis in hot organic solvents [86,89]. Ternary I-III-VI QDs prepared as core QDs exhibit poor PL QYs of less than 20% and are not stable to photon-irradiation. Surface coating of the QDs with a materials having large band gap is a suitable way to enhance the PL QY as well as the stability. The surface-to-volume ratio is higher due to their small size, as a result surface defects...
on unsaturated bonds on the surface provide non radiative decay for the photo created charge carriers.

Efficient surface passivation is required to eliminate surface defects, improve the fluorescence QY and stability. Introducing Zn or ZnS also was adjusted on optimizing optical properties of semiconductor QDs. The growth of ZnS shell layer around the surface of core enhances the PL QY of the QDs. Most researchers choose ZnS as a capping material due to its chemical properties. First, it has wide band gap (3.7eV) [90] and smaller ionic radius which forms good band alignment with III-VI QDs. Second, its ZnS blends structure allows to eliminate the surface trap states and avoid leakage created charge carriers in ternary core QDs.

Jaffe et al. [91] studied the electronic structure of Cu based ternary semiconductors i.e. CuAlS$_2$, CuGaS$_2$, CuInS$_2$, CuAlSe$_2$, CuGaSe$_2$, and CuInSe$_2$. They had studied their band structure, density of states, electronic charge densities. Gunin [92] prepared ternary I-III-VI compounds having chalcopyrite structure. He had synthesized CuInS$_2$, CuInSe$_2$, CuGaS$_2$, and AgInS$_2$ in aqueous or acetonitrile medium and studied their quantum size effect. Parkin et al. [93] synthesized CIS nanocrystals by reflux method. Lu et al. [94] synthesized CIS by using solvothermal method. They had used CuCl, In and S as reagents and synthesized at 200°C. Malik et al. [95] synthesized tri-octylphosphate oxide(TOPO)-capped CuInS$_2$ nanoparticles. They had carried out the reaction in two steps. In first step, InCl$_3$ and CuCl were reacted in tri-octylphosphate (TOP) and injected into TOPO at 100°C. In second step, at an elevated temperature of 250°C tri-n-octylphosphineselenide (TOPSe) was added to the reaction mixture. Li et al. [96] prepared CuInS$_2$ nanowhiskers and nanoparticles at 180°C. Jiang et al. [97] synthesized CIS nanorods by solvothermal route by setting the reaction temperature at 280°C for 48 hrs. In 2001, Xiao et al. [98] prepared CIS nanorods with length 400 to 450 nm and diameter 20 to 25 nm by using hydrothermal technique at 180°C for 15 hrs. Castro et al. [99] synthesized CIS and CIS$_2$ chalcopyrite compounds by thermal decomposition of single source precursors but the resultant nanocrystals were not small enough to exhibit quantum efficiency. Therefore, in 2004 they had slightly modified their method and successfully synthesized CIS nanoparticles of size 2-4 nm at 200°C. Grisaru et al. [101] prepared CuInTe$_2$ (CIT) and CuInSe$_2$ (CISE) nanoparticles of tetragonal structure by microwave assisted method. Wei et al. [102] synthesized cube shaped CIS nanoparticles by using a simple wet chemical process. Vittalet et al. [103] produced wurzite or sphalerite CIS nanocrystals by the thermal decomposition of single source precursors. Dutta and Sharma [104] synthesized CIS nanoparticles by thermolysis of precursors at 196°C. Gardner et al. synthesized 3-5 nm CIS nanoparticles using microwave irradiation of single source precursors. The other material or compounds which comes in I-III-VI group are CuGaS$_2$(CGS), AgInS$_2$ (AIS), AgInSe$_2$ (AlSe) and AgGaS$_2$ (AGS) [105].

Tang et al. [106] synthesized Zn doped AIS QDs and studied that the incorporation of Zn dopant leads to increased structure stability and crystallinity. Shapiro et al. [107] synthesized dulfatrine CuInS$_2$ nanoparticles from the photolytic decomposition of a molecular single source precursor. Vittal et al. [108] synthesized orthorhombic AgInSe$_2$ nanorods at 185°C. Monodisperse AgInS$_2$ nanoparticles were successfully synthesized by Tian and co-workers using direct thermal decomposition of the single-source precursor in a bisulfurant system [109]. Later on, the same group synthesized polyhedral shaped AgInS$_2$ nanocrystals from dual-source precursors [110]. Wang et al. [111] demonstrated one-pot synthesis strategy for the preparation of ternary nanocrystals. They had synthesized AgInS$_2$ nanocrystals having different morphologies by controlling the reaction conditions. They had also synthesized CuInS$_2$ and AgInSe$_2$ nanocrystals by replacing the precursors and provided a controlled way for the synthesis of ternary nanocrystals. Xie et al. [112] synthesized CIS and AIS nanocrystals of size range from 2 to 20 nm at 180°C by using hot injection method. Norako et al. [113] had prepared CIS nanoparticles of size 6.9 nm by hot injection method at 180°C. Park and Kim [114] synthesized CIS/ZnS core/shell QDs by cation exchange reaction with quantum yield of 65% and studied their photoluminescence. Meng et al. [115] synthesized high luminescent AIS nanocrystals by one pot reaction and study their photocatalytic and photovoltaic applications. Li et al. [116] synthesized CIS nanocrystals and deposited CdS or ZnS layers over the nanocrystals to improve their quantum efficiency. They demonstrated that the method used by them produced nanocrystals with high yield and revealed the time resolved spectroscopic studies of core and core/shell nanoparticles. They had found that the core-only particles were promising candidates for PV applications, while core/shell particles could be used in displays, solid state lighting, and bioimaging.

Allen et al. [117] achieved tunable emission from red to near-infrared region by varying the stoichiometric ratio of CISe QDs via hot injection method. They had demonstrated that the synthetic method can be used for the synthesis of ternary I-III-VI nanomaterials. They had also synthesized AIS$_2$ QDs with emission in orange to red region of spectrum by following same synthesis method and replacing the precursors with AgI and InI$_3$. Pan et al. [118] synthesized monodispersed CIS ternary nanocrystals by using mixed precursors via a hot injection method. They had synthesized NCs having zincblende to wurzite structure by varying the precursor concentration and combination of capping agent. Later, [118] the same group synthesized (CIS)/(ZnS) alloyed nanocrystals by using wet chemical method. They have achieved tunable band gap by varying the composition of the precursors. Zhong et al. [119] synthesized CISe nanocrystals with controlled stoichiometric composition and size. The synthesized CISe/ZnS core/shell QDs exhibited 26% quantum yield with excellent PL properties. Thus, they had used this material as light emitter for the fabrication of electroluminescent devices. Omata et al. [120] synthesized CISe/ZnSe core/shell QDs with PL quantum yield of 16%. They have demonstrated that there is blue shift in the emission after the deposition of ZnSe that can be attributed due to the partial alloying of CISe core with the ZnSe shell around their interfaces. They also synthesized ZnSe/CISe inverted core/shell QDs and studied their optical properties. Yarema et al. [121] synthesized high yield CISe QDs with precisely controlling the shape and size. These QDs exhibited bright luminescence and quantum yield of 50%-60%

after coating ZnS and ZnSe shell suggesting their use in bioimaging, biolabeling and lighting applications. Deng et al. [122] had proven that increasing Zn on AlSe system also effect the blue shift on PL emission, which improve QY up to 50%. Separately, Deng [123] also investigated optical properties of AlSe deposited with ZnS shell. As a result, QY of AlSe QDs can reach to 40% with emission tunable from 700 to 820 nm. Xie et al. [112] demonstrated that one layer of ZnS over the CIS core improved the PL QY to 30%. Jang et al. [124] synthesized CIS core and passivated with two layers of ZnS shell and fluorescence QY was upgraded to 92%. These double layer shell of ZnS shows a remarkable blue shift from 660 to 559 nm. The stoichiometric ratios of Cu:In showed high photostability under UV illumination both in toluene and aqueous solutions. Tang et al. [106] compared Zn doped AIS with pure AIS QDs and indicated that the incorporation of Zn dopant shows higher structure stability and crystallinity. Uehara et al. [125] enhanced the fluorescence of CIS QDs by introducing crystal defects through a highly Cu defect composition of CIS. This enhanced fluorescence was due to large number of donor or acceptor defects required for donor-acceptor pair recombination (DAP) of excited charge carriers. Hamanaka et al. [126] reported the decrease in PL QY in CIS QDs and associated it with deep surface trapper than DAP. These reports open a way for researchers to construct efficient approach for modifying the surface of ternary core QDs. Dai et al. [85] and Kameyama et al. [127] reported utilization of non-injection or pyrolysis method on synthesis of AIS and AlSe QDs. However, even prepared with precisely stoichiometric composition, difficulty to get QDs with single crystal structure commonly occurred on this type. For instance, in synthesis of AIS proposed by Dai et al. [85], they obtained the AIS QDs with tetragonal and orthorhombic crystal along with cubic AgInS$_2$ (Fig. 11). Torimoto et al. [128] synthesized ZnS-AIS nanoparticles with broad and tunable emission spectrum which makes them competent for use in solar cells and photocatalysis.

![Figure 11](image-url)

**Fig. 11.** Absorption spectra (a) and PL emission spectra (b) of AIS nanoparticles. The excitation wavelength in PL spectra was 430 nm. (c) XRD patterns of AIS nanoparticles. (Reprinted with permission from ref 85. Copyright 2012 The Royal Society of Chemistry).

Later on, in 2010 the same group synthesized ZnS-AIS nanoparticles with photoluminescence intensity of 80% using thermal decomposition of precursor at 180°C [129]. Ogawa et al. synthesized orthorhombic AIS by thermolysis and studied their photoluminescence properties [130]. Tang et al. [131] synthesized AIS–ZnS heterodimers by using hot injection method that showed tunable emission from green to red and also showed two photon fluorescence (TPF) properties. Hamanaka et al. [132] studied the photoluminescent properties of AIS QDs. They had investigated that the PL emission is attributed due to the donor-acceptor (DA) pair recombination. They suggested that the large Stokes shift observed in PL band was due to the strong interactions between the carriers which were trapped at the donor or acceptor levels. Dai et al. [85] synthesized AgInS$_2$ (AIS) semiconductor nanoparticles by the thermal composition of single-source precursors. They had varied the ratio of Ag content and achieved
maximum quantum yield of 70% with tunable emission from visible to near infrared region.

Chung et al. used carbon dots and Zn-doped AgInS2 nanocrystals for the fabrication of white LED. Carbon dots were prepared by thermal carbonization of citric acid whereas AgInS2 nanocrystals were prepared by thermal decomposition at a low temperature. Then, they fabricate a white LED by combining a UV LED with the carbon dots and Zn-doped AgInS2 NCs which exhibit warm color temperature with an excellent color rendering index 95, and verify its potential for use in solid state lighting [133]. Wang et al. prepared Ag doped ZnSe ternary quantum dots by following aqueous route and refluxed at 100°C. They exhibited high stability, good biocompatibility, low cytotoxicity, and broad tunable emission scope suggesting their use in optical coding, white LED, and bio-imaging [134].

Kameyama et al. [127] synthesized nanocrystals of ZnSe-AlSe and used them for the fabrication of quantum-dot-sensitized solar cells (QDSSC) with maximum energy conversion efficiency of 1.9%. Langevin et al. [135] synthesized air stable AlSe NCs through thermolysis followed by anion exchange reaction. These pyramid shaped NCs with quantum yield of 21% showed emission in range of 800 to 1300 nm. Lu et al. [136] fabricated white LED by using AIS/ZnS quantum dots. They had deposited the red emitting QDs over the yellow emitting phosphor layer and achieved white light having color temperature of 3500 K, luminous efficiency of 72.1 lm/W and color rendering index of 85.

(b) Other Ternary CNCs

There are other ternary nanocrystals that include II-III-VI semiconductor compounds (II:Zn, Cd or Hg; III: Al, Ga or In; VI:S, Se or Te). These compounds have gained much interest in few decades due to their tunable emission, large full width half maxima (FWHM), long decay lifetime and large stokes shift. These properties make these compounds suitable for various applications.

Chen et al. successfully synthesized porous ZnInS microspheres by thermal solution method at low temperature. The resultant product showed enhanced visible light photocatalytic activity. This low cost method was highly suitable for large scale industrial production [137]. Kim et al. [138] fabricated white light emitting diode (WLED) by using different composition of Mn-doped Zn0.75Cd0.25S quantum dots (QDs). They obtained the color rendering index (CRI) of 62 and CIE chromaticity coordinates of (0.529, 0.459) and (0.316, 0.264) for QD-based orange and white LEDs respectively. Gou et al. synthesized ZnInS, CuInS, and CuInSe nano- and microstructures via a facile solution-based route. They synthesized ZnInS nanotubes and nanoribbons by solvothermal method and ZnInS hollow microspheres by hydrothermal method in the presence of cetyltrimethylammoniumbromide (CTAB) or poly(ethylene glycol) (PEG) surfactant [139].

Shen et al. synthesized ZnInS nanocrystals by hydrothermal method and studied their photocatalytic activity for hydrogen evaluation [140]. Cao et al. [141] synthesized high-quality Cu-doped Zn-In-Se nanocrystals by using hot injection method. They have achieved composition dependent tunable emission by varying Zn to In ratio with PL quantum yield of 38%. They fabricate LED by using these QDs with maximum luminance of 320 cd m⁻² and a luminous efficiency (LE) of 0.97 cd A⁻¹. Peng et al. in 2013 synthesized ternary hexagonal ZnInS with controlled size and shape. Nanocrystals of different sizes i.e. 2.1 nm to 9.2 nm had been obtained at different reaction temperature. They demonstrated the photocatalytic activity of the ZnInS nanocrystals for degradation of methyl orange in visible region [142].

Xu et al. [143] synthesized Mn-doped ZnCd1-xS QDs using nucleation-doping approach and studied their optical properties. They have found that the emission arises from Mn-ion states and defect states. The achieved quantum yield is 29% that can be improved by decreasing non-radiative centers through growing ZnS shell. Goutam De et al. [144] fabricated films of Cu-doped Cd0.5Zn0.5S alloyed QDs incorporated with organically modified SiO2 and capped with P123. The concentration of QDs was 40 mol% higher than equivalent SiO2. The fabricated films show high luminescence and covers the entire visible region.

Zeng et al. [145] fabricated Ag-doped ZnCdS:ZnS core/shell nanocrystals at 250°C. Then, they treated the synthesized NCs with Se to prepare Ag doped ZnCdSSe nanocrystals. The size, shape, crystallity and optical properties were unchanged. Ouyang et al. [146] synthesized CdS and CdZnS quantum dots by hot injection method and observed upconversion luminescence. In 2008, they synthesized alloyed ZnCdS quantum dots by non-injection method [147]. The cubic shaped ZnCdS nanoparticles with intense luminescence and narrow full width at half maxima can be considered as blue emitting material for various applications. Shen et al. [148] fabricated quantum dots based LED by using ZnCdS:ZnS core/shell quantum dots. They had attained maximum external quantum efficiency >10% with low turn on voltage of ~2.6V. They had suggested that further optimizing the capping ligands and the electron and hole transport layers help in improving the quantum efficiencies.

Chen et al. [149] synthesized water soluble Mn-doped ZnCdS/ZnS core/shell quantum dots. They have synthesized these QDs by using an aqueous route and achieved quantum yield of 20% after coating with ZnS shell. These quantum dots having tunable emission could be used for preparing QD-based LEDs. Wang et al. [150] synthesized monodisperse zinc indium sulfide (ZIS) alloy and tuned the optical properties by varying the indium content in the nanocrystals. Li et al. [151] in 2016 synthesized transition metal ion (Mn²⁺ or Cu⁺) doped ZnInS/ZnS core/shell quantum dots (QDs) and studied their photoluminescence properties in solution as well as in films forms. Their results showed them efficient for photodetection device.Shen et al. [152] in 2011 synthesized transition metals (Cr, Mn, Fe, Co) doped ZnInS and studied their photocatalytic activity. Sarkar et al. [153] in 2011 synthesized Cu-doped ZnInS ternary alloy nanocrystals by precipitation and surface cation-exchange method and achieved emission over visible region. The synthesized ultrasmall nanocrystals had versatile characteristics which makes it useful in biomedical, solar cell, and LED applications. Yuan et al. [154] fabricated white light emitting diodes by using Cu-doped ZnInS/ZnS core/shell nanocrystals. The QDs synthesized by hot injection method showed emission in deep red to green region. The QDs based LEDs showed high photoluminescence quantum yields.
(PL QYs), low scattering and good colour saturation as compared to traditional phosphors based LEDs. They obtained device parameters as high CRI up to 96, luminous efficacy of 70–78 lm/W, CCT of 3800–5760 K, and CIE colour coordinates (0.3388, 0.3337). Zhang et al. [155] synthesized Cadmium-Free Cu-Doped ZnInS/ZnS Nanocrystals by non-injection method and achieved quantum yield of 85%. They studied the electroluminescent devices performance of the fabricated QD-based light-emitting diodes with a maximum luminance of 220 cd m$^{-2}$, low turn-on voltages of 3.6 V. Yuan et al. [156] synthesized manganese and copper co-doped Zn-In–S Quantum Dots and used them for the fabrication of white LED. The fabricated device showed color rendering index of 95, luminous efficacy of 73.2 lm/W, and color temperature of 5092 K. Huang et al. [157] in 2016 synthesized dual emissive Ag,Mn:ZnInS Quantum dots and used to fabricate temperature sensor device. The emission achieved at 485 nm was attributed to Ag-related and another vacancy while emission at 585 nm was attributed to the Mn$^{2+}$ transition from $4T_{1}$ to $4A_{2}$. 

Jiang et al. [158] synthesized Cu doped ZnInS/ZnS nanoparticles with tunable emission in the whole visible region by aqueous method. They had found them biocompatible to use in biomedical applications. Xuan et al. [159] synthesized Ag-doped ZnInS/ZnS nanocrystals and achieved quantum yield of 43.5%. They used the synthesized nanocrystals for bioimaging by changing their media to aqueous via ligand exchange. Yuan et al. [160] studied the luminescence properties of Cu-doped ZnInS/ZnS core/shell nanocrystals. They discussed the thermal quenching mechanism and used the synthesized quantum dots for fabrication of white LEDs. 

Yakoubi et al. [161] synthesized high quality Cu: CdZnS/ZnS core/shell QDs with a PL quantum yield (PLQY) of 42% and studied their optical and structural properties. These stable and water dispersible QDs could be potential candidate for biolabelling applications. Yadav et al. [162] synthesized Cu-doped Cd$_{1-x}$Zn$_{x}$S quantum dots by using chemical co-precipitation method and studied their structural properties. These QDs exhibit hexagonal crystal structure with size in the range of 2-12 nm. Levchuk et al. [163] synthesized highly efficient Mn-doped ZnCdS:ZnS core/shell QDs with high reproducibility in large quantities by using a non-injection method. They have deposited these QDs on top of a commercial monocrystalline silicon (mono-Si) solar cell and significantly enhanced its external quantum efficiency (EQE) to 12% and thus power conversion efficiency (PCE) also increases by nearly 0.5 percentage points. Thus, the production costs of mono-Si solar cell modules reduced by around 3.33%.

C. Quaternary semiconductor nanocrystals

Quaternary QDs have also been exploited to fabricate white LEDs. Zhang et al. successfully developed quaternary Cu–Zn–In–S (CZIS) QDs, and studied the optical performance of quantum dots. A new synthetic approach was exploited to fabricate various homogeneous CZIS QDs by varying reaction temperature and the ratios of the precursors, respectively. These as-prepared QDs exhibited not only broad PL spectra with emission color tuned from the visible to the NIR region but also high QY (over 70%) without coating the wide band gap shell materials. Further studies revealed good thermal stability, where the CZIS QDs did not show any obvious variation in emission intensity upon heating to 150°C [164]. 

Song et al. [165] recently developed color-tunable Cu-In-Ga-S (CIGS) QDs and CIGS–ZnS core–shell QDs by adjusting the ratios of In:Ga to apply in white LEDs. The EL spectra of CIGS–ZnS QD based LEDs with different In:Ga ratios had been collected at 20 mA input current. With increasing In content in the CIGS–ZnS QDs, the conversion efficiency of the LED increased from 74 to 96%. Lately, Liu et al. [166] in 2015 synthesized quaternary ZnCuInS$_2$/ZnSe/ZnS QDs (ZCIS/ZnSe/ZnS) to exploit them for fabricating white LEDs. Despite this recent progress, there are still remaining challenges: compared to the binary QDs, the ability to control the size, size distribution, and optical properties of quaternary QDs is still poor.

McDaniel et al. [167] synthesized CISOs QDs and used them as a sensitizer to enhance the photovoltaic performance of solar cell. The cation exchanged QDs increased the photovoltage and photocurrent in QDs based solar cell. They had achieved 3.45% of photovoltaic efficiency without optimizing the absorption in the film by refining QDs loading. Panthani et al. [168] synthesized CuSe/ZnS core/shell QDs with bright photoluminescence and quantum yield of 40%. These QDs did not contain toxic heavy metal and showed emission in red to near infrared region. Thus, they can be considered as promising candidates for biocompatible medical imaging.

Jie Wang et al. [169] synthesized ZAISOs QDs with tunable emission and QY 30%. Xiang et al. [170] in 2014 synthesized AIS and AZIS QDs using hot injection method. The AIS QDs showed tunable emission with PL QY of 62% with Ag/Zn ratio of 1/4. They had blue shifted emission spectra by introducing Zn ions i.e. AZIS. They fabricated QD based LEDs and studied their electroluminescence properties. Yoon et al. [83] in 2015 synthesized eco-friendly Zn–Ag–In–S (ZAIS) and Zn–Cu–In–S (ZCIS) QDs by hot injection method with quantum yield of 61% and 53%, respectively. These QDs were used to fabricate white down-converted light-emitting diodes (DC-LEDs). The various parameters were calculated with luminous efficacy (LE) of 31.2 lm/W, color rendering index (CRI) of 97, color quality scale (CQS) of 94 with warm white at a correlated color temperature (CCT) of 3500 K. Chung et al. [171] in 2011 fabricated white LED by using highly luminescent deep red emitting CuInS$_2$/ZnS (ZCIS) alloy nanocrystals. They had tuned the emission wavelengths by varying the Zn concentration in alloyed nanocrystals and studied their optical properties. They had calculated the device parameters of the fabricated white LED with luminous efficacy of 4.15 lm/W and CIE chromaticity coordinates of (0.3891, 0.3357) at 20 mA. 

Peng et al. [172] synthesized Mn-doped ZnCuInS/ZnS core/shell nanocrystals that showed band edge emission along with dopant emission. Mn-doped ZnCuInS/ZnS nanocrystals with various band gaps had been prepared with controlled experiment conditions. Zhang et al. [164] synthesized quaternary CuInS$_2$/ZnSe nanocrystals of various sizes and composition by simply changing reaction temperature and the ratio of the precursors. The results demonstrate its efficient use in
fabrication of quantum dots based LEDs. Kim et al. [173] produced quaternary Zn-Cu-Ga-S (ZCGS) QDs by alloying Zn into Cu-Ga-S (CGS) host QDs. The variation in composition gives tunable emissions in the azure-to-blue-range with quantum yields in the range of 78–83%. Their efficient properties makes them practically applicable for the fabrication of optoelectronic devices.

Therefore, the basic understanding of the photophysical process that takes place in colloidal nanocrystals is important. The emission mechanism for ternary and quaternary nanocrystals is different from binary nanocrystals. The composition variation in ternary nanocrystals enables stokes shifted tunable emission spectra. Moreover, deposition of shell of higher band gap material enhances the intensity and stability of nanocrystals. Thus, alloying, doping and functionalization of nanocrystals makes them efficient for use in different applications.

III. CONCLUSION

In this review, the binary, ternary and quaternary semiconductor nanocrystals have been investigated in detail and their role for the development of light emitting diodes have been discussed. This review focuses on the development of nanocrystals. The various synthesis methods to synthesize nanocrystals and their optical properties have been explored. The inclusion of dopant element into the crystal structure induces new energy states in the band gap of nanocrystals, thus changes the emission pathway. Ternary nanocrystals have been used in wide range of applications as compared to binary nanocrystals. They show tunability in their photoluminescence behaviour, broad absorbance band, large stoke-shift. These crucial optical properties facilitates their use in light emitting diodes, solar cells, bioimaging etc. The significant work has been done to improve the quantum yield of nanocrystals to make their efficient use in various applications. The white light emitting diodes have been successfully fabricated by using these nanocrystals. Although, researchers have got success in obtaining good device parameters. However, the research is going on to obtain even better external quantum efficiency of fabricated LEDs along with other device parameters.

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